

Correlation of hole mobility, exciton diffusion length, and solar cell characteristics in phthalocyanine / fullerene organic solar cells^(a)

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要旨 金属フタロシアニン (MPc, $M = \text{Fe, Co, Ni, Cu, Zn}$ and H_2) およびフラーレン (C_{60}) を積層させたヘテロ接合型有機太陽電池において, MPc のホール移動度および励起子拡散長を太陽電池特性と比較した。その結果, ZnPc を除く MPc を用いた太陽電池では短絡電流がホール移動度と線形関係を示した。また, それぞれの MPc について励起子拡散長を理論計算によって求めたところ, 励起子拡散長は短絡電流と比例関係を示し, 両者が強い相関関係にあることが確認された。

Abstract The authors investigated heterojunction organic solar cells composed of different metal phthalocyanines (MPcs, $M = \text{Fe, Co, Ni, Cu, and H}_2$)/fullerene (C_{60}) and compared the solar cell characteristics with the field-effect hole mobilities (μ_h) and exciton diffusion length (L_{ex}) of the different MPcs. They observed that the short circuit current (J_{SC}) is linearly dependent on the μ_h of the MPcs, except for ZnPc. They also estimated the L_{ex} of the MPcs by creating a line of best fit using the action spectra of the external quantum efficiency in the solar cells and found that J_{SC} is closely correlated with the L_{ex} of the MPcs.

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Organic solar cells (OSCs) have recently been proposed as a key for energy supply because they cover a large area, are inexpensive and produce flexible solar cells. By incorporating multilayers⁽¹⁾ and bulk heterojunctions⁽²⁾, the energy conversion efficiency (η_p) of OSCs was gradually increased $\sim 5\%$ ⁽³⁾. However, the η_p is still low for practical applications and has been anticipated to clarify the controlling factors required to obtain a higher conversion efficiency. Also, since only a limited number of organic materials can function in OSCs, establishing more effective material designs and expanding the number of materials required to produce OSCs are needed. In photovoltaic processes, solar cell characteristics are expected to have strong correlation with not only the carrier transport but

also exciton diffusion. Since only excitons that reach a donor-acceptor interface, i.e., a p - n heterojunction interface, are dissociated into holes and electrons, a longer exciton diffusion length can contribute to generating more charge carriers⁽⁴⁾. Further, the dissociated holes and electrons have to move to opposite electrodes by using an internal built-in electric field. Therefore, organic materials having a long exciton diffusion length (L_{ex}) and high carrier mobility (μ) are desirable for improving the η_p . Since the μ and L_{ex} are strongly dependent on the molecular stacking, a high μ generally could cause a long L_{ex} . In fact, a high η_p has recently been achieved by using pentacene⁽⁵⁾ and tetracene⁽⁶⁾ as a p -type semiconductor and fullerene (C_{60}) (Ref.7) as an n -type semiconductor. These materials are known to have

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high carrier mobility and a rather large L_{ex} due to their high molecular packing nature⁽⁸⁾⁻⁽¹⁰⁾.

In this work, we fabricated heterojunction OSCs having different metal phthalocyanines (*MPcs*) as a *p*-type semiconductor and C_{60} as an *n*-type semiconductor and compared their solar cell characteristics with the hole mobility (μ_h) and L_{ex} of the *MPcs*. To date, the *MPcs* have been widely used in *p*-type semiconductor layers in OSCs, and in this experiment, we used six Pc derivatives: FePc, CoPc, NiPc, CuPc, ZnPc, and metal-free phthalocyanine (H_2Pc). The CuPc, H_2Pc and ZnPc derivatives are well established as *p*-type semiconductors in OSCs⁽¹¹⁾⁻⁽¹³⁾, and the FePc, CoPc, and NiPc derivatives were used for comparison. The central metals of *MPcs* the Fe, Co, Ni, Cu, and Zn, are transition metals, and they all have different magnetic properties derived from the *3d* orbital. Therefore, the *MPcs* have different molecular orbital levels, but the atomic radii and the weights of the central metals are quite similar⁽¹⁴⁾.

The OSCs were fabricated on a pre-cleaned indium tin oxide (ITO)-coated glass substrate. The organic layers were grown by using thermal evaporation in a vacuum less than 3×10^{-3} Pa. Before the organic layers were deposited, all *MPcs* were purified more than two times by using thermal gradient train sublimation⁽¹⁵⁾. We used C_{60} (Frontier Carbon Corp.) as received, and no further purification was performed. A 40-nm-thick *MPcs* layer was deposited on the substrate, followed by a 30-nm-thick C_{60} layer as the *n*-type semiconductor and a 10-nm-thick bathocuproine (BCP)

layer to block excitons. A 50-nm-thick Ag film was then deposited through a shadow mask to define a circle-shaped active area with a diameter of 500 μm . After deposition, we measured the current density-voltage (*J-V*) characteristics using a source-measurement system (Advantest R6243) under AM1.5G illumination with a power of 100 mW/cm² produced by a solar simulator (Oriel 66901).

Organic field-effect transistor structures were also fabricated by depositing *MPcs* on a Si/SiO₂(300nm) substrate having comb-type electrodes composed of Cr (1nm)/Au (40nm) on the surface. The channel length and width were 25 μm and 4 mm, respectively, for 19 pairs. Before the *MPcs* were deposited, the Si substrates were cleaned by using acetone in an ultrasonic cleaner and by boiling them in isopropanol, and were then coated with 1,1,1,3,3,3-hexamethyldisilazane as a hydrophobic layer. After fabrication, we immediately measured the drain current-voltage (I_D - V_D) characteristics using a semiconductor parameter analyzer (Agilent 4155C) in a vacuum less than 1×10^{-2} Pa.

Figure 1(a) shows the *J-V* characteristics of OSCs using different *MPcs*, and Table I lists the OSC characteristics. Each OSC showed clear photovoltaic characteristics, i.e., open circuit voltage (V_{oc}), short circuit current (J_{sc}), and fill factor (FF), demonstrating that η_p is apparently dependent on *MPcs*, ranging from 0.019% to 1.1%.

First, we will discuss the *MPcs*' dependence on J_{sc} . The origins of the J_{sc} are rather complicated because the J_{sc} is

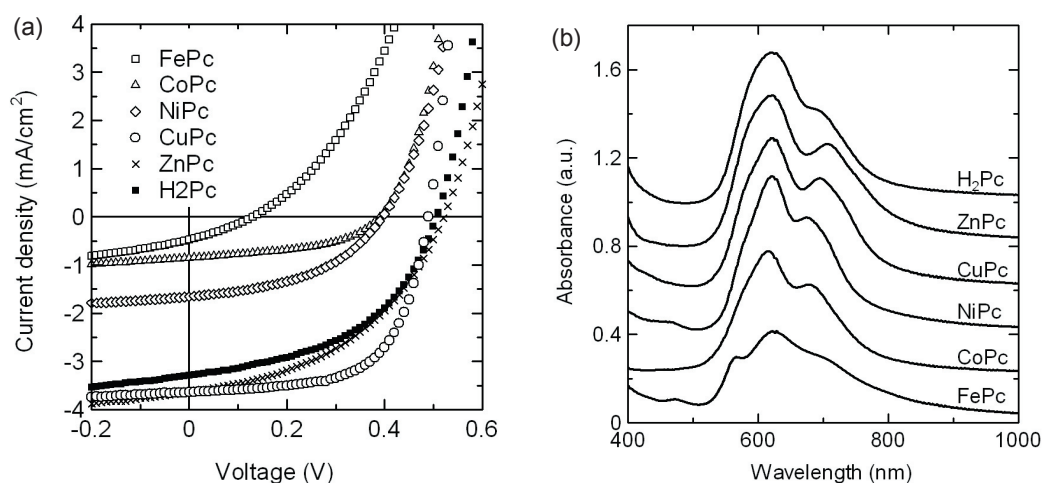


FIG. 1 (a) Current density (*J*) – voltage (*V*) characteristics in solar cells with different metal phthalocyanines (*MPcs*). (b) Absorption spectra of FePc, CoPc, NiPc, CuPc, ZnPc, and H_2Pc deposited films with thickness of 100 nm.

determined by various factors: (i) the absorption coefficient of the organic layers, (ii) the effective area of the p - n junction, (iii) the absorbed light intensity distribution inside the cell, (iv) the diffusion efficiency of the excitons, (v) the transport efficiency of the charge carriers, and (vi) the collecting efficiency of the charge carriers. The absorption intensities of the MPc s are corrected depending on the absorption spectra, as shown in Fig. 1(b). Second, we can assume that the p - n junction width in each cell is the same. Third, since the imaginary refraction indices of the MPc s are similar in these derivative^{(16),(17)}, the absorbed light intensity distribution can be assumed to be the same. Fourth, since the difference of the highest occupied molecular orbital (HOMO) levels between the MPc s is so small, as listed in Table I, and the energy barriers between the ITO anode and MPc s are comparable, we can neglect the correcting efficiency of holes at the interfaces between MPc s and the

anode. After all, the J_{SC} can be controlled by the factors of the diffusion efficiency of the excitons (iv) and the transport efficiency of the charge carriers (v). As J_{SC} is the sum of $J_h + J_e$, where J_h is the hole current and J_e is the electron current, we can assume that J_e is independent of the type of MPc s. Thus, J_h and J_{SC} can be expressed by using the linear sum of the drift current and the diffusion current,

$$J_h = n_h \mu_h \nabla U + kT \mu_h \nabla n_h, \quad (1)$$

$$J_{SC} = J_h + J_e, \quad (2)$$

where n_h is the concentration of holes, U is the electrical potential, k is Boltzmann's constant, and T is the temperature. Based on Eqs. (1) and (2), J_{SC} is proportional to μ_h . Therefore, J_{SC} can be primarily controlled by carrier mobility, i.e., the μ_h , of the MPc s in these devices.

Figure 2(a) shows a comparison of the J_{SC} and the μ_h of the MPc s obtained by using their FET characteristics.

TABLE I Summary of OSC characteristics having different MPc s as a p -type semiconductor layer, HOMO/LUMO levels of MPc s, and the L_{MPc} and η_{CMPc} that we estimated.

MPc	η_p (%)	V_{OC} (V)	J_{SC} (mA/cm ²)	FF (%)	HOMO/LUMO (eV)	L_{MPc} (nm)	η_{CMPc}
	0.019	0.13	0.48	30	4.78 / 3.32	1.0	0.14
CoPc	0.16	0.39	0.85	48	4.93 / 3.37	1.6	0.17
NiPc	0.30	0.40	1.7	45	4.68 / 3.44	9.0	0.21
CuPc	1.1	0.49	3.6	62	4.82 / 3.13	15.4	0.33
ZnPc	0.84	0.52	3.6	44	4.75 / 3.28	15.0	0.33
H ₂ Pc	0.80	0.51	3.3	48	4.86 / 3.21	11.9	0.34

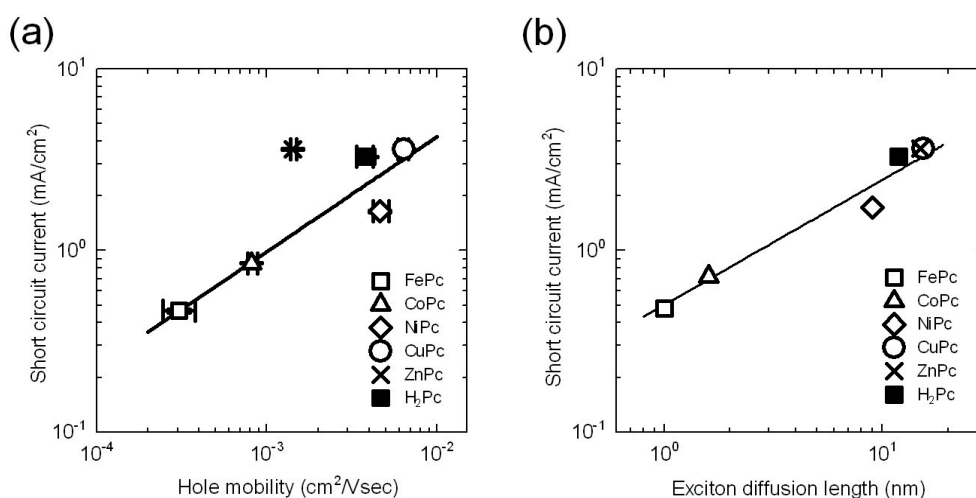


FIG. 2 (a) J_{SC} vs field effect hole mobility of metal phthalocyanines. The solid line is linearly fitted, except for ZnPc. (b) J_{SC} vs exciton diffusion length of MPc s.

We found that the J_{SC} has a rather good linear relationship with the field effect μ_h , except for ZnPc. The ZnPc showed a large deviation from the linear relationship, suggesting the presence of other controlling factors in the J - V characteristics.

Next, we will discuss the exciton diffusion length of the MPc s. Since charge carriers can be generated from exciton dissociation in the OSCs⁽¹⁸⁾, the exciton diffusion can significantly influence the carrier-generation process in the OSCs. Measuring the exciton diffusion length of MPc s directly is difficult because there is no photoluminescence in the MPc s' neat films. Therefore, we estimated the exciton diffusion length using the action spectra of the external quantum efficiency (EQE) in the OSCs. The EQE is defined as the ratio of the number of collected carriers to the number of incident photons and is expressed as follows:

$$EQE(\lambda) = \frac{hc}{e\lambda} \cdot \frac{J_{SC}(\lambda)}{P_{ph}(\lambda)}, \quad (3)$$

where h is Planck's constant, c is the speed of light, e is the elementary electric charge, λ is the wavelength, and P_{ph} is the power of incident light. In the EQE action spectra shown in Fig. 3(a), two broad peaks due to MPc and C_{60} can be identified. In the long-wavelength component longer than $\lambda \sim 550$ nm, which originates from the MPc absorption, the EQE showed a large difference, while the absorption spectra of MPc s are quite similar.

When estimating the exciton diffusion length, we considered that excitons should follow the steady state diffusion equation as follows⁽¹⁹⁾:

$$L_j^2 \frac{\partial^2 n_j(x)}{\partial x^2} - n_j(x) + \tau_j \cdot \frac{Q_j(x)}{hc/\lambda} = 0, \quad (4)$$

where L_j is the diffusion length of excitons, $n_j(x)$ is the density of excitons, τ_j is the lifetime of excitons, and $Q_j(x)$ is the time-averaged absorbed optical energy. Suffix j indicates an organic layer, e.g., n_{MPc} and $n_{C_{60}}$ are the exciton densities inside the MPc and C_{60} layer. The $Q_j(x)$ is proportional to the $|E(x)|^2$ ⁽¹⁹⁾, where $E(x)$ is the optical electric field that can be calculated by using the transfer matrix method. In solving diffusion equation [Eq. (4)], we assumed an n_j of zero at the p - n junction interface and an $\partial n_j / \partial x$ of zero at the ITO/ MPc and C_{60} /BCP interface as the boundary conditions. Although the MPc and C_{60} interface may be inhomogeneous due to the diffusion of both materials, we assumed the clear interface as a first approximation in Eq. (5). Consequently, we can obtain the density of excitons that reach the p - n junction, using the following equation:

$$n_j(x_0) = \frac{L_j^2}{\tau_j} \left. \frac{\partial n_j(x)}{\partial x} \right|_{x=x_0}, \quad (5)$$

where x_0 is the position of the p - n junction. We assumed that the excitons that arrive at the p - n junction are dissociated

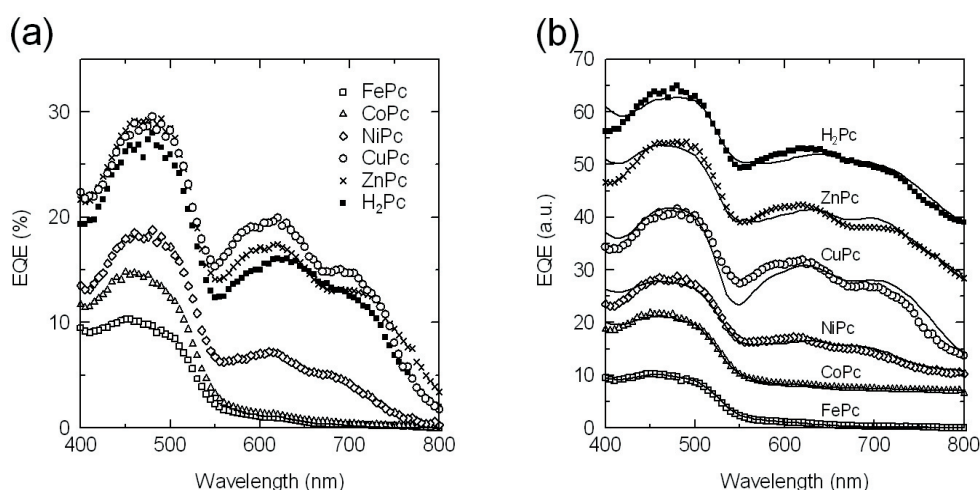


FIG. 3 (a) Action spectra of external quantum efficiency (EQE) in solar cells with MPc s. (b) Experimental and calculated EQE action spectra in solar cells using MPc s. The solid lines are calculated EQEs. Exciton diffusion lengths of MPc s determined by best fitting are L_{FePc} of 1.0 nm, L_{CoPc} of 1.4 nm, L_{NiPc} of 9.1 nm, L_{CuPc} of 15.7 nm, L_{ZnPc} of 14.7 nm, and L_{H2Pc} of 11.9 nm.

into holes and electrons with the probability of η_{ED} . In addition, since a great number of holes and electrons accumulate near the interface, we should consider the recombination probability, η_R . Therefore, the number of holes and electrons contributing to J_{SC} is expressed as follows:

$$n_{CC} = \eta_{ED} \eta_R (n_{MPc} + n_{C_{60}}) = \eta_C (n_{MPc} + n_{C_{60}}), \quad (6)$$

where n_{CC} is the number of photo induced charge carriers, and η_C is the exciton-to-charge carrier conversion efficiency, defined as $\eta_C = \eta_{ED} \eta_R$. Since η_{ED} and η_R depend on the combination of the donor and acceptor materials, we can assume that η_C depends on MPc , and η_C can be described as η_{MPc} . We assumed that all the photo generated charge carriers are collected at the anode and the cathode. The EQE is defined as n_{CC} normalized by the number of incident photons. Therefore, the EQE consists of L_{MPc} , $L_{C_{60}}$, and η_{MPc} . We fitted the calculated EQE to the experimental data by varying both the L_{MPc} and η_{MPc} , as shown in Fig. 3(b), where a $L_{C_{60}}$ of 40 nm was assumed⁽⁴⁾. As the result, we obtained η_{MPc} and $\eta_{C_{60}}$, as shown in Table I. Based on these estimations, we found an almost linear correlation between J_{SC} and L_{MPc} , as shown in Fig. 2(b). All MPc s fall along the line of best fit, i.e., strongly suggesting that the exciton diffusion length of the MPc s controls the J - V characteristics of the OSCs. Therefore, we can expect that the OSC characteristics can enhance a considerably higher η_P by using organic semiconductor materials with a long L_{ex} .

We found that the V_{OC} and FF also tend to increase with μ_h . The maximum V_{OC} of 0.52 V in ZnPc is four times larger than that of the minimum V_{OC} of 0.13 V in FePc. There is no straightforward correlation between the V_{OC} and the energy gap between the lowest unoccupied molecular orbital (LUMO) level of the acceptor (C_{60}) and the HOMO level of the donor (MPc s) since the differences in the HOMO levels of the MPc s are much smaller than those of V_{OC} . Gregg and Hanna proposed that the V_{OC} is controlled by a chemical potential gradient of the OSCs⁽¹⁸⁾. The chemical potential gradient is equivalent to the carrier density gradient and would depend on the carrier mobility. Based on this idea, V_{OC} is dependent on μ_h .

References

- (1) C. W. Tang, Appl. Phys. Lett. **48**, 183 (1986).
- (2) M. Hiramoto, H. Fujiwara, and M. Yokoyama, Appl. Phys. Lett. **58**, 1062 (1991).
- (3) J. Xue, S. Uchida, B. P. Rand, and S. R. Forrest, Appl. Phys. Lett. **85**, 5757 (2004).
- (4) P. Peumans, A. Yakimov, and S. R. Forrest, J. Appl. Phys. **93**, 3693 (2003).
- (5) S. Yoo, B. Domercq, and B. Kippelen, Appl. Phys. Lett. **85**, 5427 (2004).
- (6) C. W. Chu, Y. Shao, V. Shrotriya, and Y. Yang, Appl. Phys. Lett. **86**, 243506 (2005).
- (7) P. Peumans and S. R. Forrest, Appl. Phys. Lett. **79**, 126 (2001).
- (8) O. D. Jurchescu, J. Baas, and T. T. M. Palstra, Appl. Phys. Lett. **84**, 3061 (2004).
- (9) R. W. I. de Boer, T. M. Klapwijk, and A. F. Morpurgo, Appl. Phys. Lett. **83**, 4345 (2003).
- (10) Y. Kubozono, T. Nagano, Y. Haruyama, E. Kuwahara, T. Takayanagi, K. Ochi, and A. Fujiwara, Appl. Phys. Lett. **87**, 143506 (2005).
- (11) J. Xue, S. Uchida, B. P. Rand, and S. R. Forrest, Appl. Phys. Lett. **84**, 3013 (2004).
- (12) T. Taima, M. Chikamatsu, Y. Yoshida, K. Saito, and K. Yase, Appl. Phys. Lett. **85**, 6412 (2004).
- (13) K. Suemori, T. Miyata, M. Yokoyama, and M. Hiramoto, Appl. Phys. Lett. **86**, 063509 (2005).
- (14) M-S. Liao and S. Scheiner, J. Chem. Phys. **114**, 9780 (2001).
- (15) H.J. Wagner, R.O. Loutfy and C.K. Hsiao, J. Mater. Sci., **17**, 17 (1982).
- (16) Z. T. Liu, H. S. Kwok, and A. B. Djuricic, J. Phys. D **37**, 678 (2004).
- (17) O. D. Gordan, M. Friedrich, and D. R. T. Zahn, Thin Solid Films **455/456**, 551 (2004).
- (18) B. A. Gregg and M. C. Hanna, J. Appl. Phys. **93**, 3605 (2003).
- (19) L. A. A. Pettersson, L. S. Roman, and O. Inganäs, J. Appl. Phys. **86**, 487 (1999).

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